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## ABSTRACT

[Several phen](http://www.sciencedirect.com/science/journal/00406031)omenological models (including simple models by Ozawa and Kissinger–Akahira–Sunose isoconversional method) have been used to compare of two lignin-based novolac-type phenolic resins with those from a commercial nov 45 wt% phenol is substituted by a sulfonated kraft lignin, an important reduction in th is obtained. This behaviour has been attributed to the incorporation of an extra amoun groups in the formulation, as they are present in important quantities in the original groups. KAS isoconversional model shows that the rise in viscosity derived from lignin intr moderate change in the limiting stage from a kinetic to a diffusion regime, while cond which are favoured by the abundance of lignin hydroxymethyl groups, acquire high i system cure. Finally, competition with other mechanisms initiated at high tempera high conversion grades for all cases.

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### **1. Introduction**

Phenolic (PF) resins were the first completely synthetic plastics to be created and, since then, they have acquired huge importance through their use in many applications in the automotive, construction, electric and aerospace industries. Among all applications, their use as adhesives in wood panel products, which is derived from their excellent bonding behaviour, is one of the most relevant. This application accounted for up to 55% of the whole PF consumption of the United States in the year 2000 [1].

Due to increasing environmental restrictions, in the last decades there has been a huge research initiative to reduce the relatively high formaldehyde emissions associated with these resins. A possibility that is receiving much attention is the introduction of lignin (L) in the resin formulation as a substitute for part of the petrochemical derived components. L is a polyphenolic macromolecule present in the cell wall of plants and is, with the exception of cellulose, the most abundant renewable natural resource on Earth. Depending on the original source and pulping method used for its extraction, physicochemical characteristics of L and thus its suitability in phenolic formulations vary noticeably [2].

Due to the fact that they are thermosetting systems, it is critically important to control the curing process of PF resins by which they are transformed from viscous liquids to elastic gels (gelation) and finally to glassy solids (vitrification). From an of view this is absolutely necessary for obtaining properties. The study of cure kinetics contributes edge of the complex process governing the curin helps to improve the quality of final products.

Cure kinetics can be analyzed by using mee nomenological models. Mechanistic models sho as they are based on the balance of all the sp consumed during the curing process. In the resins this is extremely complicated due to the of curing reactions, involving the formation of of intermediate compounds such as benzoxazine imines, amides, imides and many others  $[1,3-5]$ . Mechanism els become even more complex when the resin are L-based. Phenomenological models based on  $\epsilon$ found to be more appropriate for these cases  $[6-8]$ different methodologies can be used for the dete kinetic parameters. Some direct methods, e.g., B [9] employ a single dynamic differential scanning  $\epsilon$ scan to calculate these parameters, while others l and Kissinger [12] models, use multiple dynam same objective. Isoconversional models are anoth model and provide more accurate information as changes in the activation energy throughout the These models are especially interesting for student curing process of phenolic resins.

In this work, two simple phenomenologic els (Ozawa and Kissinger) as well as an isocon

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<span id="page-1-0"></span>dependent:

$$
\frac{d\alpha}{dt} = A \exp\left[\frac{-E_a}{RT}\right] f(\alpha) \tag{1}
$$

with *A* the Arrhenius frequency factor, or pre-exponential factor, and *E*<sup>a</sup> the activation energy per mol.

For non-isothermal conditions, when the temperature is raised at a constant heating rate  $(\beta)$  the integral form of the single step kinetic equation can be expressed as

$$
g(\alpha) = \int_0^{\alpha} \frac{1}{f(\alpha)} d\alpha = \frac{A}{\beta} \int_{T_0}^{T} \exp\left[\frac{-E_a}{RT}\right] dT
$$
 (2)

As has been widely reported [7,10–12,18–20], the use of different approximations to solve the temperature integral give rise to Ozawa (Eq. (3)) and Kissinger (Eq. (4)) expressions where the activation energy is related to the heating rate and the temperature at the peak of the exothermic curve  $(T_p)$  observed by DSC.

$$
-\ln \beta = 1.0516 \left(\frac{E_a}{RT_p}\right) - A'\tag{3}
$$

$$
-\ln\left(\frac{\beta}{T_{\rm p}^2}\right) = \frac{E_a}{RT_{\rm p}} - \ln\left(\frac{A^1R}{E_a}\right) \tag{4}
$$

A plot of  $-\ln \beta$  *vs*  $1/T_p$  or  $-\ln (\beta/T_p^2)$  *vs*  $1/T_p$  gives a straight line from which activation energy can be determined which activation energy can be determined.

Isoconversional models are based on the assumption that the activation energy remains constant for a given conversion grade regardless of curing temperature. This aspect has been confirmed for the curing process of both PF [16,17] and LPF [7,21,22] resins. In this work, KAS model [23] has been used for the determination of kinetic parameters of LPF resins cure over the whole conversion range.

[Conversion](#page-4-0) grade is defined as

$$
\alpha = \frac{\Delta H_{\alpha}}{\Delta H_0} \tag{5}
$$

where  $\Delta H_{\alpha}$  is the heat released at a given conversion value, cor-<br>responding in dynamic scaps to a single temperature, and AH. responding in dynamic scans to a single temperature, and  $\Delta H_0$ <br>the total heat released at the end of the reaction Considering the the total heat released at the end of the reaction. Considering the expression described above for Kissinger simple model and using the definition of conversion as a function of the enthalpy (Eq. (5)), the following expression is obtained for [KAS mo](#page-2-0)del:

$$
-\ln\left(\frac{\beta}{T_{\alpha}^{2}}\right) = \frac{E_{\alpha}}{RT_{\alpha}} - \ln\left(\frac{A_{\alpha}^{1}R}{E_{\alpha}}\right)
$$
(6)

For a given heating rate, each conversion grade is associated with a single temperature  $(T_\alpha)$  and thus the corresponding activation energy  $(E_\alpha)$  can be determined from the slope of the straight line obtained by plotting  $-\ln(\beta/T_\alpha^2)$  *vs*  $1/T_\alpha$  in the same way as for simple models. [models.](#page-4-0)

been partially substituted by L at two different proportions (25 and 45 wt%) have been synthesized, as reported  $p$ procedure follows the typical scheme used for the industry, with a previous  $L-P$  mixing step prepolymers, as well as the reference PF resi with 10 wt% of HMTA.

#### *3.3. DSC measurements*

DSC measurements were carried out in a 821e module previously calibrated with high zinc standards. Experiments were conducted working with medium pressure pans and 5-10 different heating rates (4, 8, 12, 16 and  $20^{\circ}$ C/m in a temperature range from 30 to 300 $\degree$ C, though the range at which the curing reaction takes place. enthalpy  $(\Delta H_0)$  and temperature at the peak<br>curve  $(T_1)$  of the curing reactions were recorder curve  $(T_p)$  of the curing reactions were recorde

### **4. Results and discussion**

Fig. 1 shows the thermograms obtained by D LPF-45 at different heati[ng](#page-5-0) [rat](#page-5-0)es  $(4, 8, 12, 16$  and  $T<sub>p</sub>$  values corresponding to these curves are  $T<sub>p</sub>$ These data have been used for the determination parameters through the application of Ozawa and Kissinger simples models and, in a second approach, KAS isocor all cases, correlation coefficients  $(R^2)$  higher than obtained.

As seen in Fig. 1, both L-modified systems under tive curing processes at lower temperatures o commercial PF. This behaviour reveals that th in the formulation induces a positive effect, p accessibility of L reactive positions have as th tributed homogeneously within the prepolyr  $[25]$  reported a s[imilar](#page-2-0) [b](#page-2-0)ehaviour for a phenolic 30 wt% of lignosulfonate which was attributed that this compound has over the curing proces

### *4.1. Ozawa and Kissinger simple models*

Ozawa [10,11] and Kissinger [12] models ha calculation of activation energies derived from of PF, LPF-25 and LPF-45 resins. Those kinetic pa determined representing  $-\ln \beta$  $-\ln \beta$  $-\ln \beta$  *vs*  $1/T_p$  (Ozawa  $-\ln(\beta/T_p^2)$  *vs*  $1/T_p$  (Kissinger model, Fig. 3) according to Eqs. (4) Begults are reported in Table 2 and in all c (4). Results are reported in Table 2 and in all c good reproducibility.

As can be seen in Table 2, LPF system contain an  $E_a$  value close to that of PF, while this parameter  $E_a$ an important decrease when the substitution

<span id="page-2-0"></span>



**Fig. 3.**  $-\ln(\beta/T_p^2)$  *vs*  $1/T_p$  plottings according to Kissi

resole resin. Obtained results suggest that an imp L hydroxymethyl groups have taken part in the o LPF-45, making it proceed easier and, thus, reduc energy required. LPF-25 would not show this be

### **Table 2**

Activation energies and regression factors obtained with Ozaw models



**Fig. 1.** DSC thermograms at different heating rates obtained for (a) PF, (b) LPF-25 and (c) LPF-45.

behaviour indicates that the curing process proceeds much easier in LPF-45 system. Gabilondo et al. [17] observed an important decrease in the *E*<sup>a</sup> values for a resole resin when F/P molar ratio was set to high values, which they believed to be caused by an activation effect from the excess of  $-CH<sub>2</sub>OH$  groups over phenolic rings. This kind of group is also present in the original L structure and it is well known that it can react with the rest of the components in the formulation in the same way as  $-CH<sub>2</sub>OH$  groups in a conventional

tion energy of 200 kJ/mol. However, this great difference is due to the fact that Borchardt–Daniels method tends to overestimate that parameter  $[20]$ . On the other hand, Pérez  $[22]$  determined that the substitution of 30 wt% of P with an ammonic lignosulfonate resulted in a decrease of the curing activation energy (104 kJ/mol), showing the same tendency reported in this work for the 45 wt% L-substituted resin.

# *4.2. KAS isoconversional model*

Simple models are useful tools that allow studying the curing process of a thermosetting resin in a global manner, but they provide us with very little information on the evolution of such reactions. Related to this, isoconversional models analyze the curing processes in more detail as they determine a single activation energy for each conversion  $(E_\alpha)$ . These models have been success[ful](#page-4-0)ly used for the kinetic analysis of several thermosetting resins [17,19,20,27–33]. Some of these works deal with lignosulfonatemodified resole [20–21] or novolac [22] resins, and reveal the validity of [isocon](#page-5-0)versional principle in LPF resins.

Considering conversion grade as defined in Eq. (5) it is possible to use heat flow values determined by DSC to obtain the evolution of conversion with temperature. Fig. 4 shows  $\alpha$  vs T curves at different heating rate[s](#page-4-0) [for](#page-4-0) [e](#page-4-0)ach system. By recording  $T_\alpha$  from these curves and applying KAS model, the evolution of apparent  $E_\alpha$  throughout the conversion range can be represented. The shape of  $E_\alpha$  function in this kind of curves can give valuable insight on the change in reaction steps [33,34].

Fig. 5 shows  $E_\alpha$  *vs*  $\alpha$  curves obtained from KAS model for the different systems analyzed. In general, changes in apparent  $E_\alpha$  are related to the existence of multiple reactions or changes in the limiting stage [33–35]. It is well known that PF cure involves a set of parallel reactions including HMPs formation, condensation as well as different crosslinking chemistries [1,33,36]. Also the presence of water formed during hardening may turn the process even more complicated, contributing to reversible reactions and delaying diffusion control by plasticizing PF network [33,37]. This sometimes leads to a wavy shape of  $E_\alpha$ , as reported for PF resole resins by Wang et al. [33], which remarks the complexity of PF cure kinet[ics. As](#page-4-0) can be seen in Fig. 5, rather smooth  $E_\alpha$  functions were obtained for both modified and unmodified systems. This can be in part attributed t[o hom](#page-1-0)ogeneous crosslinking processes but it also has to be considered that KAS is less sensitive than other methods to changes in  $E_\alpha$  [33].

For the reference PF resin the  $E_\alpha$  value obtained in this work remains constant (122 kJ/mol) almost throughout the entire process, indicating that it is dominated by a single reaction type [34]. The basic cure mechanism in phenolic resins is based on two main reaction types, namely addition and condensation reactions [1,33,36]. As the former is necessary for cure to proceed, since no hydroxymethyl groups are present in the novolac structure at the



Fig. 4. Evolution of conversion degree with temperature out of (a) PF, (b) LPF-25 and (c) LPF-45 resins at different heat.

end of polymerization, addition mechanism is prevailing one in this case. This also indicates ymethyl groups that are being created react w rings, what is favoured in PF by the high prop of activated free ring positions. A similar beh curing rate is determined by a single step was razzuoli et al. [38] when applying an isoconv non-stoichiometric curing of an epoxy system of amine.

<span id="page-4-0"></span>

**Fig. 5.** Evolution of apparent activation energy with conversion degree according to KAS isoconversional model for the curing process of ( $\bullet$ ) PF, ( $\vartriangle$ ) LPF-25 and ( $\triangledown$ ) LPF-45 resins.

The initial increase shown in the  $E_\alpha$  function of PF at the early stages of the cure process fits with that observed by Wang et al. [33] for a high molecular weight PF resole resin and assigned to competitive addition and condensation reactions. On the other hand, the progressive increase at high conversion grade ( $\alpha$  > 0.8) is related to the simultaneous action of those two mechanisms with contribution of new processes initiated at high temperature [39,40]. Among them, thermal decomposition and rearrangement of methylene ether linkages into methylene bridges, which is known to occur above 170 °C [1,41], are the most likely mechanisms involved.

When 25 wt% P is substituted with L, the  $E_\alpha$  function suffers a slight downward dependence until  $\alpha$  = 0.8. This function shape has been traditionally assigned to a change in the limiting stage from a kinetic to a diffusion regime [34,42], which in LPF-25 resin may be due to the increase in viscosity caused by L introduction; this aspect was reported in a previous work [24]. The same effect was described by Vázquez et al. [43] for curing of tannin-modified PF adhesives. Also the initial quick decrease of  $E_\alpha$  in the  $\alpha$  region 0–0.5 has been linked to a mechanism of viscosity-related diffusion by different authors [44–46]. However, the general decrease in the apparent activation energy of LPF-25 system is not very pronounced (<7%) in  $\alpha$  range 0.05–0.8, suggesting that viscosity has only a moderate effect on the curing process of this resin. The opposite behaviour is found again at  $\alpha$  > 0.8, where  $E_{\alpha}$  suffered a sharp increase showing the same tendency as reference PF. It is concluded that in LPF-25 the same mechanisms as those commented for PF, as well as new ones promoted by the presence of L, are taking place.

The  $E_\alpha$  curve for LPF-45 system remains more or less invariable during the initial steps of the curing process, though a slightly wavy shape is found when it is viewed in detail. In this case the effect of higher viscosity (downward dependence) is thought to be counteracted by competitive condensation reactions (upward dependence) caused by the abundance of L hydroxymethyl groups.Moreover, this latter effect would be the reason for the  $E_\alpha$  increment in the  $\alpha$  range [0.35](#page-5-0)–0.65; within these limits, unmodified PF only presents addition reactions and at the same time corresponding temperatures [are s](#page-5-0)till too low for decomposition or rearranging reactions, so the competitive mechanism that causes  $E_\alpha$  increase can only be related to L reactive groups. The final sharp increase at high conversion values will be again related to the multiple parallel reactions mentioned above, with increased contribution of those originated by L.

lignin at a 45 wt% level an important reduction energy is obtained while there is only small effect stitution. This behaviour has been attributed to of an extra amount of hydroxymethyl groups in as they are present in important quantities in t structure. KAS isoconversional model shows th tions dominate the curing process in reference the rise in viscosity derived from lignin introd moderate change in the limiting stage from a  $\mathbf k$ sion regime. Condensation reactions, favoured by lignin hydroxymethyl groups, acquire high relevan tem cure. Finally, competition with other mechanisms at high temperature is reported at high conversion g

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